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Report:

High local Pu concentrations and fission gas release in the mixed uranium oxide-plutonium oxide (MOX) fuels irradiated to high burn-ups significantly affect the fuel performances. On the whole, the manufacture of MOX fuel consists in UO_2 and PuO_2 powders co-crushing, milling, pelletizing and sintering. An evolution of the manufacture route giving rise to a microstructure optimization of nuclear fuel pellets could allow us to improve the fuel performance.

Pu should be dispersed as homogeneously as possible in a $(U,Pu)O_2$ solid solution by an enhanced powder process. In this aim, several doping agents have been studied. Encouraging results were reported for coarse-grained UO_2 by the addition at low concentration of Nb₂O₅, MgO, Al₂O₃, V₂O₅, TiO₂ or Cr₂O₃. In the case of Cr₂O₃, the first results for MOX show an increase on grain size and, above all, an increase of the Pu spreading. Pu distribution and grain size are both key parameters to reach high burn-ups.

Under the temperature and atmosphere conditions used during sintering, Cr_2O_3 is particularly interesting because of its property to give rise to a liquid phase (eutectic $Cr_s+Cr_3O_{4s} \rightarrow CrO_1$) which usually helps grain growth. Moreover, the possible presence of Cr^{3+} in the fluorine lattice of $(U,Pu)O_2$ could increase point defect concentration and enhance Pu diffusion. A specificity of the Cr-doping in presence of plutonium is the formation of a line compound the composition of which is PuCrO₃. It is the main difference between the U-O-Cr system and the Pu-O-Cr one at the temperature and atmosphere used during sintering. A PhD thesis is ongoing to determine the mechanisms allowing the microstructure enhancement of a Cr-doped MOX pellet.

The aim of this study is therefore to examine Cr_2O_3 impact on Pu distribution when the dopant is introduced up to 2000 ppm as a powder during the UO_2 -PuO₂ milling and to determine the sintering temperature dependence. We had to verify the formation of a $(U,Pu)O_2$ solid solution in the presence of a doping agent (introduced under chromium sesquioxide form) and determine the oxidation states of U and Pu. Moreover, we want to improve our knowledge of the crystallographic structure of PuCrO₃ and obtain data on oxidation states of Pu and Cr in this compound. EXAFS turned out to be a key technique to probe Pu and U environments or oxidation states and to evaluate the structural order.

During our experiment, we analyzed three samples with 2000 ppm of Cr_2O_3 , *Mic55*, *Mic67 and Mic7540*, obtained by the usual MOX powder process (*MIMAS*, *micronization of a master blend*) for a sintering temperature of respectively 1530°C, 1630°C and 1700°C. The Pu/(U+Pu) ratio of the samples was equal to 11 wt%.

A PuCrO₃ sample was analyzed. PuCrO₃ had been obtained by sintering a PuO₂-Cr₂O₃ powder mixture heated at 1700°C under argon. A small amount (less than 3 wt%) of PuO₂ and amorphous chromium oxide still remained after the thermal treatment. Prior to the experiment, PuCrO₃ crystal structure was investigated by XRD. The first results are summarized in this table :

Pbnm	a = 5.439 Å	b = 5.498 Å	c = 7.720 Å
	Cr-O = 1.97 Å	Pu-O = 2.34 - 2.66 Å	Pu-Cr = 3.19 Å

For each sample, fluorescence and transmission signals were collected at the uranium (17.126 keV) L_{III} edges, plutonium (18.056 keV) L_{III} and (22.266 keV) L_{II} edges and chromium (5.989 keV) K edge. Energy calibration of XANES data was achieved using the Cr foil (5.989 keV), Y foil (17.052 keV), Zr foil (17.998 keV) and Rh foil (23.220 keV) references located behind the second ionization chamber. As observed on these XANES spectra, the white line positions are those of the UO₂ at the uranium edge, and those of PuO₂ at the plutonium edge.

At the uranium L_{III} edge, EXAFS spectra were collected up to 15 Å⁻¹ but only up to 10.5 Å⁻¹ at the plutonium L_{III} edge and 13.5 Å⁻¹ at the plutonium L_{II} edge. The plutonium L_{III} edge limitation is due to the presence of a small amount of americium (L_{III} edge at 18.514 keV) in the samples (induced by the used plutonium isotopy). At the chromium K edge, EXAFS spectra were collected up to 12.5 Å⁻¹ and show a non-linear decrease after the edge. In MOX, whatever the sintering temperature is, the oxidation state of uranium and plutonium remains equal to +IV, the oxidation state of chromium is +III.

In PuCrO₃, plutonium and chromium are both at the oxidation state +III. These results are summarized on in figure 1 for the four samples. Cr_2O_3 XANES are added as references for Cr^{3+} .No combination of the Cr_2O_3 and the PuCrO₃ XANES allow us to reproduce correctly the XANES of the chromium in the MOX. Considering the absence of the two peaks observed in Cr_2O_3 white line and the differences with PuCrO₃ in each chromium XANES, the main chromium amount in the MOX microstructure is certainly dissolved in the fluorine lattice of (U,Pu)O₂ (after sintering between 1530°C and 1700°C). The local environment of this chromium changes gradually with the increase of the sintering temperature.



Figure 1 : XANES spectra at the chromium (5.989 kev) K edge.

Fourier transforms of EXAFS spectra collected at uranium L_{III} edges and plutonium L_{II} edges are given in figure 2 and 3. For both edges, the first peak which corresponds to the Metal-Oxygen shell, is similar for each of the MOX samples and is characteristic of a stoichiometric oxide with an O to M ratio equal to 2. The second peak which corresponds to the Metal-Metal shell, shows differences between uranium and plutonium. Indeed, in the uranium case, all samples give rise to a second peak which is close to that obtained for UO₂ powder, indicating that the uranium environment in UO₂ and U_{0.89}Pu_{0.11}O₂ is equivalent due to an UO₂ matrix effect. In the plutonium case, the large range of plutonium concentration due to the synthesis process (a high plutonium concentrated powder mixture diluted in UO₂), results in a broadening of the peak [1], [2]. The temperature moderates this phenomenon by a more even plutonium spreading in presence of the doping agent. This accounts for an increase with the temperature of the (U,Pu)O₂ solid solution homogeneity in presence of a small amount, 2000 ppm, of Cr₂O₃.



 $(k\text{-range: } 3 - 12.6 \text{ } \text{Å}^{-1}).$

Tigure 3 : Fourier transforms of the plutonium L_{II} *edge EXAFS* (*k-range: 3 – 11.9 Å*⁻¹).

During this experiment, we have been able to determine plutonium, uranium and chromium oxidation states and local environments in enhanced MOX fuel. The homogeneity of the Cr-doped $(U,Pu)O_2$ solid solution with temperature was followed through by EXAFS. Finally, it appears that the environment of chromium added in the MOX powder evolves with the sintering temperature.

[1]: A.Pieragnoli, P.Martin, M.Ripert, H.Palancher, J.Lechelle (2005) experimental report ME-939.

[2] : P.Martin, S.Grandjean, C.Valot, G.Carlot, M.Ripert, P.Blanc, C.Hennig (2007) XAS study of $(U_{1-y}Pu_y)O_2$ solid solutions accepted and to be published in Journal of Alloys and Compounds.