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Report:

The aim of this experiment was to study the possible formation of dopant cluster formation in doubly doped functional oxides by means of x-ray linear dichroism (XLD) and x-ray magnetic circular dichroism (X-MCD). The beamtime was split between soft (ID08) and hard (ID12) x-ray absorption spectroscopy. One set of samples was 3%N-codoped and undoped 2% Fe:TiO₂. Fig. 1 shows the XLD spectra at the anion sublattice of O and N K-edges, respectively. From the very similar XLD signature it can be directly concluded that the N is substitutionally incorporated in the TiO₂ host lattice. Fig. 2 shows the respective XLD at the Ti K-edge and $L_{3/2}$ -edges revealing that there are small changes at the pre-edge features of the Ti K-edge upon N-doping. The corresponding features at the $L_{3/2}$ -edges are much less pronounced. XLD and thus also XMCD at



the Fe absorption edges were impossible to be recorded, the strong Ti K-absorption because of the rutile substrate precluded any measurement at the Fe K-edge and the Fe $L_{3/2}$ -edges suffered from large background artifacts in TEY and hardly any signal in FY.

The second set of samples consists of a series of N-codoped Co:ZnO samples with a range of different N as well as Co concentrations. XMCD measurements at ID12 were limited due to the breakdown of the magnet power supply of the 17 T magnet, which made measurements of the magnetic anisotropy by M(H) curves impossible. Fig. 3 shows XLD measurements at the Co and O K-edges, respectively. The signatures reflect the overall trend also seen at the Zn K-edge (not shown) reflecting the variety of structural perfection depending on the preparation conditions. Fig. 4 shows the XLD at the N K-edge in fluorescence yield for most of the samples in Fig. 3. Comparing the XLD of the O K-edge and the N K-edge is becomes obvious that the N is not substitutionally incorporated into the ZnO lattice. The N K-edge exhibits one pronounced peak with sizable XLD which is characteristic for molecular N₂. Since both total electron yield as well as XP did not show any significant amount of N, one has to conclude that these N2 molecules may be incorporated in the ZnO matrix. Either it can be found at grain boundaries or the molecules may substitute for O vacancies which can be found due to the low amount of O in the sputter gas. Nonetheless, the XLD spectra at all four constituents do not show any indication for the formation of dopant complexes in Co:ZnO:N. In addition, the N data point towards molecular N₂ rather than substitutional doing which may also explain why N-codoping did not lead to the expected p-type conductivity in these samples yet. In contrast, the Ti K-edge of Fe:TiO₂:N shows indications of changes in the electronic structure of the host crystal upon N-doping; however, this could not be corroborated at the Fe K-edge due to the large fluorescence background of the rutile substrate.

Finally, Fig. 4 shows exemplarily two XMCD spectra and the corresponding M(H) curves at the Co K-edge in plane and out of plane, respectively. For this 1µm thick reference film the small anisotropy of the M(H) curves could be recorded with sufficient signal to noise ratio; four other samples were tried as well. The obtained five M(H) curves will be compared to those at the Co L_{3/2}-edges (not shown).

In summary, a large set of XLD and some XMCD spectra could be recorded for different doubly doped functional oxides. While for Fe:TiO₂:N substitutional incorporation of N was found the Co:ZnO:N only show signs of the incorporation of molecular N_2 into the host matrix.

