


Experiment title:

Relations between molecular mobility and density fluctuations in amorphous polymers.

Experiment number:

SC229

Beamline:

D2AM

Date of Experiment:

from: 28-Nov-96 7:00 to: 29-Nov-96 7:00

Date of Report:
Shifts:

3

Local contact(s):

E. Geissler

Received at ESRF :
2 8 FEB. 1997
Names and affiliations of applicants (*indicates experimentalists):

L. David*, A. Fake*, R. Vassoille, G. Vigier*

 Groupe d'Etude de Métallurgie Physique et de Physique des Matériaux,
 UMR CNRS 5510, Institut National des Sciences Appliquées de Lyon,
 20, av. A. Einstein

69621 Villeurbanne Cedex

E-mail: david@gemppm.insa-lyon.fr

S. Etienne*

Laboratoire de Métallurgie Physique et Science des Matériaux,

URA CNRS 155 Ecole des mines de Nancy,

Parc de Saurupt,

54042 Nancy Cedex.

Report:
Aim of the work.

The microstructural state of glasses and liquids can be characterized by the fluctuation function

$$\psi(v) = \frac{\langle N^2 \rangle - \langle N \rangle^2}{\langle N \rangle}$$

 where N is the number of electrons in the reference volume v , and $\langle \rangle$ means averaging over all volumes v in the large sample volume. For exemple, for a perfect gaz, this value is 1 and for a crystal, it is zero as soon as v is large enough. The limit $\psi(v \rightarrow \infty)$ can be measured by SAXS experiment because the intensity scattered at zero angle, $I(q \rightarrow 0)$ is proportionnal to it.

 Moreover, $\psi(v)$ can be divided into a static component ('frozen-in' fluctuations) and a dynamic component (fluctuations arising from atomic/molecular mobility).

Thus,

$$\psi(v) = \psi_{stat}(v) + \psi_{dyn}(v)$$

 In the liquid state, the value of $\psi(\infty)$ is given by statistical thermodynamics as:

$$\psi(\infty) = \psi_{dyn}(\infty) = \rho kT \chi(T)$$

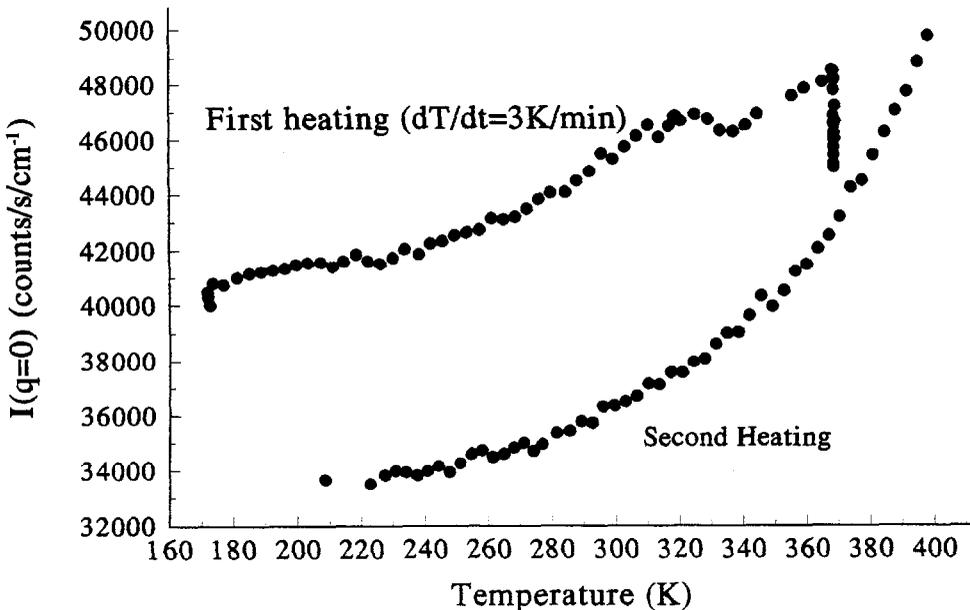
 where $\chi(T)$ is the isothermal compressibility, and ρ the electronic density.

 In the glassy state, there is no exact relation involving the properties of the glass for the prediction of $\psi(\infty)$.

 This work is a contribution to the identification of the dynamic and static components of the fluctuation function in the glassy state, as the microstructure is varied. To this goal, the effect of a thermal treatment (annealing below the glass transition temperature T_g and quenching from above T_g) was considered in our previous studies. The present experiment is devoted to the study of the effects of plastic deformation, which leads to a specific microstructural state, in terms of density fluctuations.

Effect of plastic deformation

The extrapolated values of the scattered intensity at zero angle were measured for deformed PMMA samples with applied strain in the range from 20% up to 50% in compression mode, as a function of temperature and orientation of the sample. All samples were kept in liquid nitrogen before the experiment, so that minimum microstructural reorganization occurs before the scattering experiment. The presented figure shows two successive heating runs, for a sample with applied strain close to 50%.



There is a large difference in $I(q=0)$ at low temperature between the two successive experiments. Consequently, there is a large static fluctuation contribution in the deformed glassy state. This static component also induces a shoulder below T_g , which is related to dynamic fluctuations.

When the temperature is maintained at 95°C for 2 hours, the static and dynamic contributions decrease with time, as well as deformation as shown by the evolution of the transmission factor.

As a conclusion, deformation induces fluctuations with static nature, which is not the case for a quenching treatment [1]. The disorder associated with these static fluctuations induces dynamic fluctuations which manifest themselves at higher temperature. As soon as the anelastic part of deformation is recovered, the static/dynamic fluctuations disappear. These results, collected here for the first time, are in agreement with the basic hypothesis of several models of non elastic deformation of glassy polymers, according to which the applied strain results in the reversible formation of 'sheared microdomains' [2] [3], or 'plastically transformed zones' [4] associated with an increase of energy and molecular mobility.

References:

- [1] Faivre, A. L.; David, L.; Vassoille, R.; Vigier G.; Geissler, E.; Vigier, G.; *Macromolecules*, in press.
- [2] Argon, A., S.; *Phil. Mag.*, 1973, 28, 839.
Argon, A., S.; Kuo, H., Y.; *Mat. Sci. Engng.*, 1979, 39, 110.
- [3] Perez, J.; *Physique et Mécanique des Polymères Amorphes*, Technique et Documentation - Lavoisier, 1992, ISBN 2-85206-787-0
- [4] Oleinick, E. F.; *Prog. Coll. Polym. Sci.*, 1989, 80, 140
Oleinick, E. F.; *Polym. Sci.*, 1993, 35, 1532