



<b>Experiment title:</b> <b>Phase Separation in Cuprate Superconductors:</b> <b><math>Y_{1-y}Ca_yBa_2Cu_3O_{6.98}</math> (<math>y=0.02-0.2</math>)</b>	<b>Experiment number:</b> <b>HS 533</b>	
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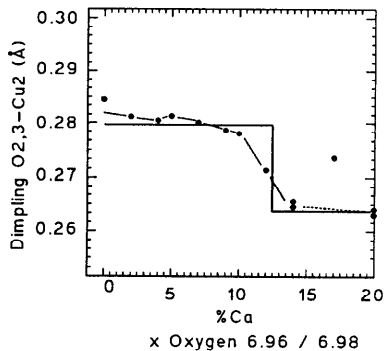
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### Report:

Access to the overdoped regime in the phasediagram of the the high- $T_c$  superconductor  $YBa_2Cu_3O_x$  may be achieved by full oxygenation of the chain system ( $x>6.95$ ) and/or by partial substitution of the separating  $Y^{3+}$  layer by  $Ca^{2+}$ . Although both types of doping, cation substitution in the separating layer, and oxygenation of the so-called insulating layer, increase the hole concentration,  $n_h$ , in the conducting  $CuO_2$  layers, they have different effects on the electronic and atomic structure of the compound. For instance it is still a matter of controversy if  $Y_{1-y}Ca_yBa_2Cu_3O_6$  ( $y>0.1$ ), i.e. the completely deoxygenated but homogenously Ca doped compound is superconducting or not. To avoid the poorly defined physical states arising from the usual dual doping (where Ca and 0 doping are considered to be equivalent), we have taken special care to ensure the highest possible oxygenation ( $x=6.96-6.98$ ) of the compund throughout the complete range accessible by Ca doping ( $y= 0-0.2$ ) Thus we have been able to extend our systematic investigations of structural and electronic phase separation phenomena far into the o v e r d o p e d regime at most precise chemical conditions.

We have measured the local atomic structure of  $Y_{1-y}Ca_yBa_2Cu_3O_6$  ( $x=6.96-6.98$ ;  $y=0-0.2$ ) by EXAFS beyond the Y-K edge ( $k < 20 \text{ \AA}^{-1}$ ) at low temperatures (20-60 K).

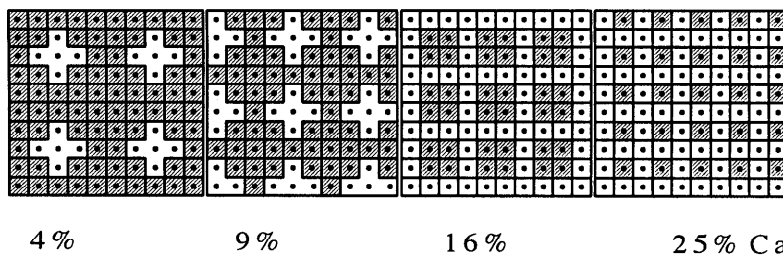
The strongly dampened EXAFS amplitudes (“amorphization”) found in some of the previous runs are most probably due to an incorrect thermal treatment. Fig.1 exhibits the dimpling of the CuO<sub>2</sub> planes vs. the Ca concentration as determined from the analysis of



**Fig.1**

Dimpling in the CuO<sub>2</sub> planes of Ca doped (Y, Ca)Ba<sub>2</sub>Cu<sub>3</sub>O<sub>6.98</sub>. The step-function indicates the behaviour expected from the critical percolation of the Ca-clusters at  $0.09 < y < 0.16$ ,  $T = 25$  K.

the two nearly collinear multiple scattering paths Y-O<sub>2,3</sub>-Ba (5) and Y-Cu<sub>2</sub>-Ba (6.2). While the average crystallographic structure (from neutron diffraction) indicates that increasing Ca content decreases the dimpling monotonously, the Y-probe of the local structure yields the dimpling to be independent on Ca-doping up to 9%. A step-like decrease by about 0.015 Å occurs around 10% Ca. Obviously the dimples in the CuO<sub>2</sub> planes located next to the Y sites remain unaffected up to a critical Ca concentration. Fig. 2 exhibits schematically (assuming the perfectly ordered case) the percolation of the Ca clusters (white) for 4, 9, 16, and 25 % Ca. Assuming that the divalent Ca atoms distort only the *nn* Y-cells, ≈10% Ca turns out to be the critical concentration for the percolation path connecting the Ca clusters and isolating the Y clusters. We conclude that the additional charge introduced by Ca<sup>2+</sup> remains essentially localized at the Ca site, and thus the compound tends to decompose into two different electronic phases. The existence of two different electronic phases is also evidenced from a recent analysis of the Meissner effect exhibiting two superconducting transitions: one with T<sub>c</sub> = 92 K independent on Ca substitution, and a second one with strongly decreasing T<sub>c</sub> upon increasing Ca content.



**Fig.2:** Percolation of the Ca clusters (white) in the Y matrix (black), schematic.