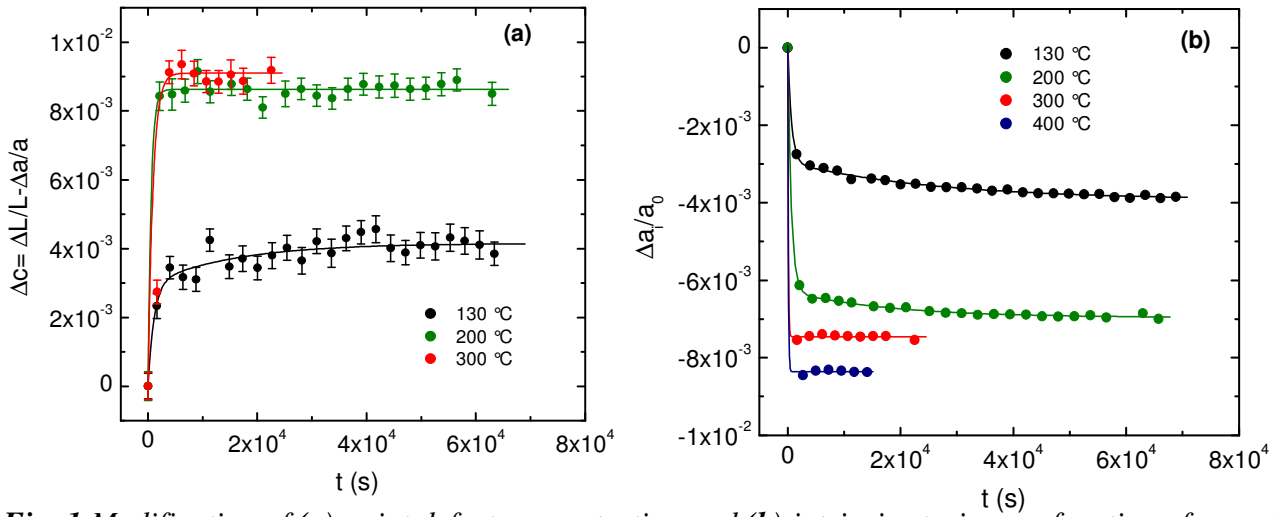
 ROBL-CRG	<b>Experiment title:</b> Determination of Point Defect Concentrations in Nano-crystalline Pt	<b>Experiment number:</b> 20-02/691
<b>Beamline:</b> BM 20	<b>Date of experiment:</b> from: 24-02-2010 to: 01-03-2010	<b>Date of report:</b> 16-08-2010
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Nano-crystalline metals are generally in a non-equilibrium state after preparation by e.g. magnetron sputtering. The aim of this work was to explore the modification of non-equilibrium point defect concentrations (PDC) and residual stresses during isothermal annealing of nano-crystalline Pt films. The out-of-plane lattice parameter,  $a$ , and the thickness,  $L$ , of a 40 nm film were measured by XRD (Bragg-Brentano) and XRR, respectively, as a function of annealing time at 130, 200, 300 and 400 °C in an in-situ vacuum furnace at  $10^{-6}$  mbar. In addition, GI-XRD ( $0.34^\circ$ ) measurements were done.

For analysis it is assumed that  $a(t, T) = a_0 + \Delta a_{th}(T) + \Delta a_{mis}(T) + \Delta a_i(t, T)$ , where  $a_0$  is the lattice parameter in the as-deposited state,  $\Delta a_{th}$  is the modification of the lattice parameter due to thermal expansion,  $\Delta a_{mis}$  due to stress resulting from thermal mismatch between substrate ( $\text{SiO}_2$ ) and Pt film and  $\Delta a_i$  due to intrinsic stress, resulting from the production of the film by sputtering. Further,  $L(t, T) = L_0 + \Delta L_{th}(T) + \Delta L_{mis}(T) + \Delta L_i(t, T) + \Delta L_v(t, T)$ ,  $\Delta L_j$  ( $j = 0, th, mis, i$ ), are the corresponding quantities of the film thickness and  $\Delta L_v$  is the thickness change due to a modification in PDC, which becomes not visible in lattice constant changes in first approximation. Note that the determined experimental resolution of the method, given by the maximum error limits of  $\sigma(\Delta L/L) < 5 \times 10^{-4}$  and  $\sigma(\Delta a/a) < 1 \times 10^{-4}$  is better than initially expected, allowing to determine changes in PDC of about  $1 \times 10^{-3}$ . The modification of PDC as function of time and temperature can be obtained by simply using the formula  $\Delta c = \Delta L_v/L_0 = (L-L_0)/L_0 - (a-a_0)/a_0$  (Fig.1(a)).



**Fig. 1** Modification of (a) point defect concentration and (b) intrinsic strain as a function of annealing time. The lines are a guide for the eye. The error limits in (b) are smaller than the dots.

As obvious, for all temperatures we observe a drastic increase of  $\Delta c$  as a function of annealing time, however limited to the first hour of annealing. This can tentatively be interpreted by a process connected with the formation of free volumes, e.g. vacancies during annealing. This is in contrast to initial assumptions, where a reduction of  $\Delta c$  is expected due to vacancy annihilation. A preliminary idea to explain the results is that vacancies are formed to counterbalance the development of stresses during heating to elevated temperatures (see below). A possible influence of an oxide layer forming at the surface is currently discarded by high resolution XPS measurements.

From analysis of  $a(t, T)$  closer information on strain modifications can be derived. It is assumed that strain modifications resulting from thermal expansion,  $\Delta a_{th}$ , and thermal mismatch,  $\Delta a_{mis}$ , are time-independent. Detailed analysis (not plotted) shows that during heating up to 400 °C we are in the thermo-elastic range, resulting in the formation of more compressive stresses. Consequently, the observed time-dependence is attributed to the modification of intrinsic strains  $\Delta a_i(t, T)$  (see Fig. 1(b)). For  $T > 200$  °C the increase in  $\Delta a_i$  takes place during the first 30 min of annealing (the actual time resolution of the experiment). However, for  $T \leq 200$ , the process slows down and becomes visible on a time scale of several hours. The extraction of characteristic time constants is in progress. Further insight into the modification of stresses was gathered from GI-XRD measurements. The discussion is omitted here due to space limitations. In conclusion, we demonstrated the feasibility of our method for the determination of PDC. For  $T \leq 200$  °C a sufficient resolution can be achieved in order to quantify strain modifications as a function of annealing time. For PDC this can also be achieved for lower temperatures and/or faster data acquisition.